

# Elucidating Mechanisms of Diffusion-Based Calcium Carbonate Synthesis Leads to Controlled Mesocrystal Formation

Johannes Ihli,\* Pieter Bots, Alexander Kulak, Liane G. Benning, and Fiona C. Meldrum\*

Aggregation-based crystal growth often gives rise to crystals with complex morphologies which cannot be generated via classical growth processes. Despite this, understanding of the mechanism is rather poor, particularly when organic additives or amorphous precursor phases are present. In this work, advantage is taken of the observation that aggregation-based growth of calcium carbonate, and indeed many other minerals, is most often observed using diffusion-based synthetic methods. By fully characterizing the widely used ammonia diffusion method (ADM)—which is currently used as a “black box”—the solution and supersaturation conditions which accompany  $\text{CaCO}_3$  precipitation using this method are identified and insight is gained into the nucleation and growth processes which generate calcite mesocrystals. This reveals that the distinguishing feature of the ADM is that the initial nucleation burst consumes only a small quantity of the available ions, and the supersaturation then remains relatively constant, and well above the solubility of amorphous calcium carbonate (ACC), until the reaction is almost complete. New material is thus generated over the entire course of the precipitation, a feature which appears to be fundamental to the formation of complex, aggregation-based morphologies. Finally, the importance of this understanding is demonstrated using the identified carbonate and supersaturation profiles to perfectly replicate  $\text{CaCO}_3$  mesocrystals through slow addition of reagents to a bulk solution. This approach overcomes many of the inherent problems of the ADM by offering excellent reproducibility, enabling the synthesis of such  $\text{CaCO}_3$  structures in large-scale and continuous-flow systems, and ultimately facilitating *in situ* studies of assembly-based crystallization mechanisms.

## 1. Introduction

Significant efforts are made to synthesize crystals with defined sizes, morphologies and structures for applications in areas as wide-ranging as pharmaceuticals, biomaterials and

J. Ihli, Dr. A. Kulak, Prof. F. Meldrum  
School of Chemistry  
University of Leeds  
Woodhouse Lane, Leeds, LS2 9JT, UK  
E-mail: cmjfi@leeds.ac.uk; F.Meldrum@leeds.ac.uk  
Dr. P. Bots, Prof. L. Benning  
School of Earth and Environment  
University of Leeds, Leeds, LS2 9JT, UK



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nanomaterials.<sup>[1]</sup> In order to achieve control over these features it is necessary to understand the mechanisms by which crystals form. At one end of the spectrum, the classical picture of crystallization from solution envisages crystal growth to occur via ion-by-ion, or molecule-by-molecule addition to an established nucleus to give a single crystal product.<sup>[2]</sup> At the other, if growth of the individual nuclei is slow, then aggregation can dominate, leading to the formation of polycrystalline particles. Importantly, such aggregation-based processes often lead to crystalline particles with unusual morphologies, such as fibers,<sup>[3,4]</sup> and “microtrumpets”<sup>[5]</sup> which cannot be accessed through classical growth processes.

Aggregation-based crystal growth is currently receiving considerable interest.<sup>[2,6–8]</sup> While the formation of polycrystalline particles based on nonoriented aggregation of nanoparticles has been recognized for a long time, single crystal formation through the oriented aggregation of precursor nanoparticles is a rather new finding.<sup>[9]</sup> One prominent example of this is the work of Banfield and co-workers which showed the formation of single crystals of titania through nanoparticle aggregation,<sup>[10]</sup> a phenomenon they also observed in iron oxide systems.<sup>[11]</sup> This has recently also been observed in the

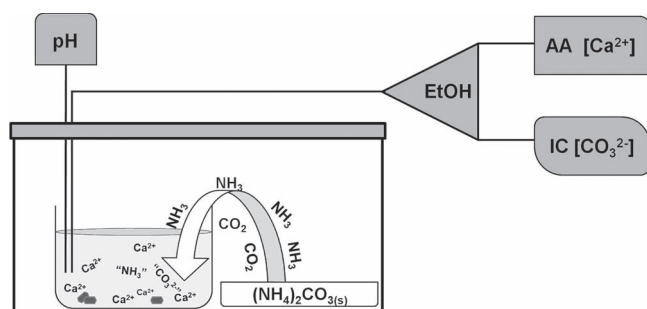
calcium sulfate system.<sup>[12]</sup> It is now well established that many crystals grow by aggregation under appropriate experimental conditions, and that a number of mechanisms can operate, ranging from the oriented aggregation of crystalline nanoparticles to the aggregation and then subsequent crystallization of amorphous nanoparticles.<sup>[6]</sup> While these processes can lead to single crystals indistinguishable from those formed by classical growth mechanisms, the crystals produced can also retain a memory of the precursor particles from which it forms. In this case, the crystal is classified as a mesocrystal, which ideally comprises a 3D array of iso-oriented single crystal particles of size 1–1000 nm.<sup>[13]</sup> The ultrastructure of such a mesocrystal clearly contributes to defining its properties, as exemplified by sea urchin spines. These calcium carbonate single crystal biomaterials have recently been classified as mesocrystals, where

the nanoparticulate substructure and residual amorphous calcium carbonate may contribute to their remarkable mechanical properties.<sup>[13,14]</sup>

Although aggregation-based growth promises the ability to produce crystals with unique morphologies and internal structures, this can only be achieved by elucidating the mechanisms by which aggregation occurs. Moving towards this goal, we here focus on calcium carbonate as an important mineral,<sup>[1,15]</sup> which grows by aggregation under appropriate experimental conditions. Our approach is based upon the common observation that diffusion-based methods, including the double diffusion,<sup>[16,17]</sup> the Kitano<sup>[18]</sup> and the ammonium carbonate diffusion method<sup>[19,20]</sup> often generate unusual crystal morphologies which cannot be accessed by other routes. This appears to be particularly true for additive-directed crystal growth, as exemplified by polymers such as poly(styrene sulfonate) or poly(4-styrenesulfonate-co-maleic acid) (PSS-co-MA), which have to-date only yielded calcium carbonate mesocrystals when using diffusion methods.<sup>[21–23]</sup>

This work characterizes the physico-chemical changes in solution which accompany CaCO<sub>3</sub> precipitation by the most widely used of these diffusion-based methods—the ammonium diffusion method (ADM)—where CaCO<sub>3</sub> precipitation is induced by exposing a solution of calcium ions to the vapor released on the decomposition of solid ammonium carbonate in a hermetically-sealed container (Figure 1).<sup>[19,20]</sup> This was achieved by performing time-resolved measurements of solution pH, carbon and calcium ion concentrations, and identifying how these are determined by key variables including the gas-liquid interfacial area, the CaCl<sub>2</sub> concentration, the initial mass of ammonium carbonate, the stirring rate and the presence of a secondary diffusion barrier. A unique insight into the ADM is therefore generated by (i) identifying the variables which principally dictate the precipitation products and (ii) determining for the first time the carbon addition rates, and the supersaturation and reaction profiles.

To demonstrate the power of this understanding, we then use the identified carbon addition rates to reproducibly precipitate CaCO<sub>3</sub> mesocrystals in the presence of PSS-co-MA, through slow addition of reagents to a bulk solution. This provides an excellent “test” of our approach. The ability to prepare such CaCO<sub>3</sub> mesocrystals using alternative synthetic methods opens the door to industrial scale and potentially one-pot syntheses,<sup>[24]</sup>



**Figure 1.** Schematic diagram of the experimental setup and methods used to determine the concentrations of aqueous Ca<sup>2+</sup> and CO<sub>3</sub><sup>2-</sup> ions; sample aliquots were removed and quenched with ethanol, the [Ca<sup>2+</sup>] was determined using atomic absorption (AA), and the [CO<sub>3</sub><sup>2-</sup>] using ion-chromatography (IC).

and will ultimately facilitate in situ studies of mesoscale assembly. Further, by identifying the solution conditions which give rise to CaCO<sub>3</sub> mesocrystals, we gain insight into their possible mechanisms of formation.

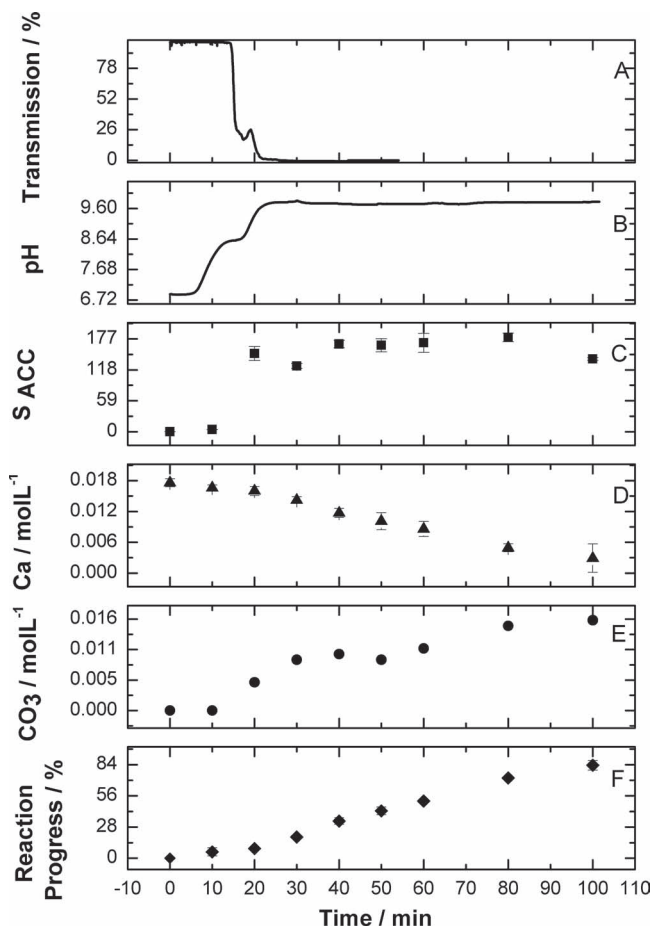
## 2. Results

### 2.1. General Description of the Ammonia Diffusion Method (ADM)

The ammonium diffusion method is characterized by two distinct steps, the first being the rapid saturation of the gas phase with CO<sub>2</sub> and NH<sub>3</sub> and the subsequent diffusion of the CO<sub>2</sub> and NH<sub>3</sub> molecules across the gas-liquid interface into the solution. In the slower, second step, the aqueous carbon dioxide reacts with the solution water to form carbonic acid, which in turn deprotonates to give both carbonate and bicarbonate ions, where the ratio of these species is defined by the solution pH. The dissolved NH<sub>3</sub>, in turn, increases the pH of the solution. In combination with the calcium ions present, a solution that is supersaturated with respect to CaCO<sub>3</sub> is generated. The equations describing the solution equilibria are given in the Supporting Information.

The ADM was initially characterized from a standard set-up using 70 mL of 25 mM CaCl<sub>2</sub>, with an air/solution surface area of 48 cm<sup>2</sup>, 3 g of uncovered (NH<sub>4</sub>)<sub>2</sub>CO<sub>3</sub> and 2.6 L free volume in the reaction chamber. Time-resolved measurements were made of the key solution variables pH, calcium, and carbonate concentration, which together yielded the carbon addition rates and underlying supersaturation profiles. Typical graphs showing the time-resolved changes in (A) turbidity (transmittance), (B) pH, (C) supersaturation, (D) calcium activity, (E) carbonate activity, and (F) crystallization progress are presented in Figure 2. The turbidity measurements show a rapid decrease at ≈15 min (Figure 2A). This is likely caused by the formation of detectable amounts of calcium carbonate precipitate, thus revealing an instrumental induction period of 15 min. The rapid drop in transmission after 15 min is associated with the formation of amorphous calcium carbonate (ACC) which subsequently transforms to vaterite and ultimately calcite, as confirmed by Raman and IR spectroscopy (Supporting Information Figure S1). The small increase in transmission at around 20 min is observed in almost all experiments, and may result from the transition of ACC to a crystalline phase.<sup>[17,25]</sup>

The turbidity measurements are also consistent with the pH data (Figure 2B). Knowledge of the solution pH is critical to understanding the ammonia diffusion method as it governs the distribution of aqueous carbonate speciation, and is required to determine the fraction of carbonate in the total carbon content measured.<sup>[26]</sup> A higher pH is associated with an increase in the concentration of carbonate at the expense of bicarbonate, and also promotes the dissolution of gaseous carbon dioxide into the CaCl<sub>2</sub> solution, through its conversion to bicarbonate or carbonate.<sup>[27]</sup> This process is reflected in the jump in supersaturation observed between 10 and 20 min (Figure 2C). Nucleation of CaCO<sub>3</sub> therefore only occurs after a critical pH/supersaturation has been reached (at pH ≈ 8.5), which marks the partial transformation of bicarbonate to carbonate. As shown



**Figure 2.** Time-resolved profiles of calcium carbonate precipitation using the ADM, with reaction conditions: 25 mM  $\text{CaCl}_2$ , 70 mL, 3 g ammonium carbonate, 2.6 L reaction chamber, no additional diffusion boundaries. The data shown are averages of three experiments, and the error bars show the standard deviation in the values. A) Transmission, B) pH, C) supersaturation, D) calcium activity, E) carbonate activity, and F) crystallization progress.

in the data, this pH value is coincident with the induction point recorded using turbidity measurements. The pH then continues to increase until it reaches a value of  $\approx 9.8$ , where it remains rather constant ( $[\text{CO}_3^{2-}]/[\text{HCO}_3^-] \approx 0.38$ ) until the crystallization is almost complete. It then decreases very slowly to a constant value of  $\approx 9.2$  after about 20 h (Supporting Information Figure S2).

These data therefore indicate that in the ammonia diffusion method, initial ACC nucleation (at the induction point of  $\approx 15$  min) occurs above an ACC supersaturation threshold of  $>10$ . Further, initial crystallization takes place under an excess of calcium ions, such that the ratio of calcium to carbonate ions is  $\approx 2.2$ . Comparison with alternative  $\text{CaCO}_3$  precipitation methods therefore shows that the ADM results in initial ACC precipitation at relatively low supersaturation levels. Indeed, precipitation of ACC using the direct precipitation method,<sup>[25]</sup> (where 1 M calcium and carbonate solutions are combined and precipitation occurs after an induction period) occurs at  $S_{\text{ACC}} > 100$ , while supersaturations of  $S_{\text{ACC}} > 30$  at pH  $> 12$  are

associated with Koga's method,<sup>[28]</sup> (where calcium and high pH carbonate solutions are combined giving immediate precipitation). These supersaturation values were calculated using Visual MINTEQ based on experimental parameters provided in these papers. As a further distinction between these different precipitation methods, the direct method and Koga's method remove up to 90% of the total precipitation stress via the initial burst of ACC formation. In contrast, in the ADM the calcium and reaction profiles tend toward classical S shapes.

Following the onset of nucleation, the calcium ion profile undergoes an extended linear decrease due to its consumption during  $\text{CaCO}_3$  precipitation (Figure 2D), while continued release of fresh ammonium carbonate vapor into the reaction chamber (which continues until equilibrium is reached after  $\approx 20$  h) supports an increase in the solution carbonate concentration (Figure 2E). Consequently, the supersaturation continues to increase after the induction point until it peaks at its maximum value of  $\approx 180$  between 60–80 min under these reaction conditions. The supersaturation then decreases only very slowly, due to the continued introduction of carbonate into the solution. High supersaturation levels, which are well above the critical value for ACC, are therefore maintained even when a significant proportion of  $\text{Ca}^{2+}$  ions have been consumed. This prolonged period of high supersaturation is a key feature of the ammonia diffusion method and would be expected to support multiple nucleation events.

## 2.2. Carbonate Addition Rate

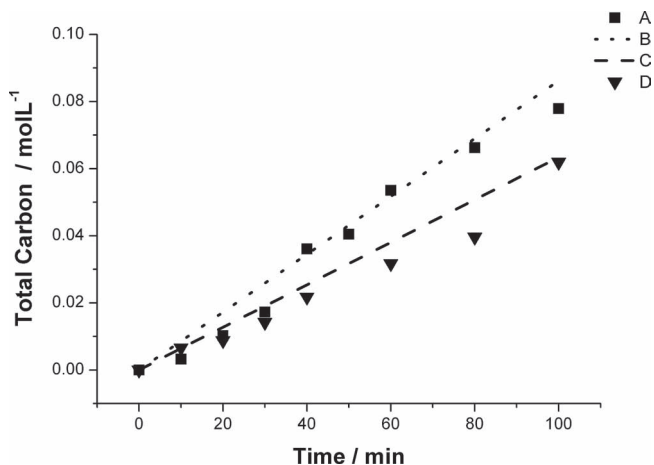
Having identified the key features of the ADM, methods were developed to produce identical results using highly reproducible titration-based methods. This was achieved based on determination of the carbon addition rate ( $d[\text{C}]_{\text{T}}/dt$ ). Here,  $[\text{C}]_{\text{T}}$  corresponds to the total inorganic carbon ( $[\text{C}]_{\text{T}} = [\text{CO}_2] + [\text{H}_2\text{CO}_3] + [\text{HCO}_3^-] + [\text{CO}_3^{2-}]$ ) added to the system, which equals the sum of carbon lost to  $\text{CaCO}_3$  precipitation and the total free inorganic carbon present in the solution ( $[\text{C}]_{\text{TS}}$ ). The total carbon which has been added to the solution at any point in time ( $[\text{C}]_{\text{T}}$ ) can be calculated from the reduction in the calcium ion concentration ( $\Delta[\text{Ca}] = [\text{Ca}]_0 - [\text{Ca}]_t$ ), as determined by atomic absorption spectroscopy, and from the total carbon content in the solution ( $\Delta[\text{C}]_{\text{TS}} = [\text{C}]_{\text{TS}t} - [\text{C}]_{\text{TS}0}$ ), as measured using Ion Chromatography, (Equation 1).

$$[\text{C}]_{\text{T}} = ([\text{C}]_{\text{TS}t} - [\text{C}]_{\text{TS}0}) + ([\text{Ca}]_0 - [\text{Ca}]_t) \quad (1)$$

The carbon addition rate i.e., the rate at which carbon dioxide diffuses into the solution ( $d[\text{C}]_{\text{T}}/dt$ ) is then obtained by differentiating Equation 1 with respect to time, to give Equation 2.

$$\frac{d[\text{C}]_{\text{T}}}{dt} = ([\text{C}]_{\text{TS}} - [\text{Ca}]) \frac{d}{dt} \quad (2)$$

The experimental data (Figures 2 and Figure 3) shows that at times up to  $\approx 100$  min (when the reaction is almost complete and the Ca concentration depleted) both the  $[\text{Ca}]$  and  $[\text{C}]_{\text{TS}}$  vary linearly with time, and that the pH is almost constant during the region of interest. The change in calcium and carbon concentration in solution can therefore be approximated using first order rate



**Figure 3.** Comparison of total carbon addition rates: A) experimental, B) theoretical C) theoretical  $k_1$  only, and D) Experimental pure diffusion no calcium chloride.

constants,  $[C]_{TS} = k_1 t$ , and  $[Ca] = k_2 t$ , where  $t$  is the experimental time in minutes and  $k_1$  and  $k_2$  are rate constants corresponding to the carbon addition rate and the crystallization rate respectively. Equation 2 can thus be expressed as Equation 3, where  $k_1 = 0.000634 \text{ mol L}^{-1} \text{ min}^{-1}$  and  $k_2 = -0.000228 \text{ mol L}^{-1} \text{ min}^{-1}$ , based on the data presented in Figure 2.

$$\frac{d[C]_T}{dt} = k_1 - k_2 \quad (3)$$

Graphs of the experimentally-obtained carbon addition profiles under the given “standard conditions” are presented in Figure 3, where (A) is the experimentally-obtained addition rate, (B) is theoretical, calculated using the derived rate constants  $k_1$  and  $k_2$ , (Table 1) (C) is theoretical, calculated using  $k_1$  only and (D) is experimental, based on diffusion of carbon dioxide into water rather than calcium chloride solution. The curves show that there is indeed a linear increase in the total carbon content until crystallization is virtually complete (at  $\approx 100$  min). Further, this comparison demonstrates that the analysis made in Equation 3 well-describes the reaction in the first 100 min under these standard conditions.

### 2.3. Influence of Reaction Variables

In describing the use of the ammonia diffusion method in the literature, authors typically place great weight on precisely defining variables such as the initial amount of ammonium carbonate, the calcium concentration, the solution surface area, the presence of further diffusion barriers and solution agitation. The effects of these variables were therefore investigated in order to determine their importance on the total carbon addition rate.

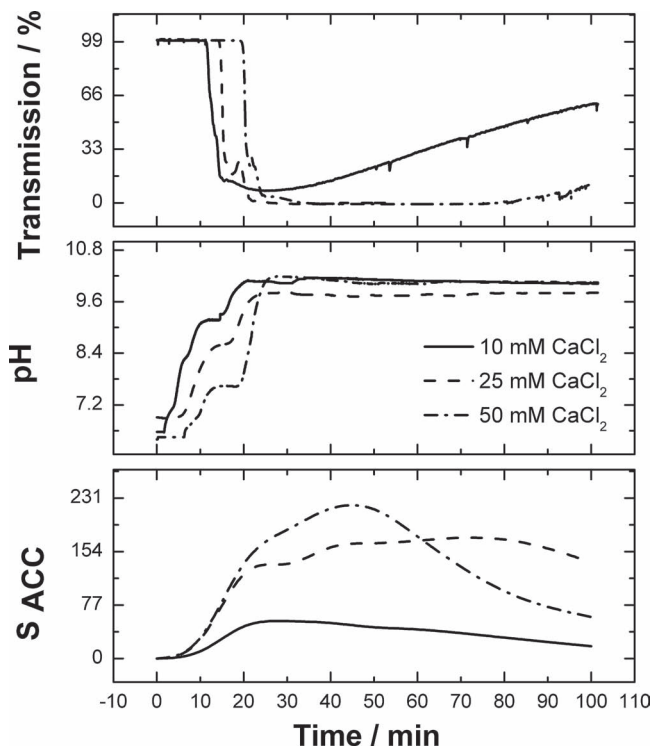
#### 2.3.1. Initial Mass of Ammonium Carbonate

Provided that the initial amount of solid ammonium carbonate was in excess, the precise mass used in an experiment had no influence on the precipitation profile. This is shown in the total

**Table 1.** Summary of ADM crystallization parameters under different experimental conditions, where ( $k_1$ ) is the carbon accumulation rate in solution, ( $k_2$ ) is the crystallization rate, ( $d[C]_T/dt$ ) the total carbon addition rate, ( $d[CO_3]/dt$ ) the carbonate addition rate, ( $t_{ind}$ ) the induction time and ( $S_{ind}$ ) is the supersaturation with respect to ACC at the induction point.

Crystallization	$k_1$	$k_2$	$d[C]_T/dt$	$d[CO_3]/dt$	$t_{ind}$	$S_{ind}^a$	Polymorph Distribution <sup>b)</sup>	
	[mm/min]	[mm/min]	[mm/min]	[mm/min]	[min]		Primary	Secondary
Base Condition <sup>a)</sup>	$0.634 \pm 0.045$	$-0.228 \pm 0.007$	0.849	0.138	15	31	Calcite	Vaterite
Mass $(NH_4)_2CO_3$								
5 g	$0.650 \pm 0.012$	$-0.224 \pm 0.038$	0.864	0.140	16	30	Calcite	Vaterite
1.5 g	$0.654 \pm 0.095$	$-0.243 \pm 0.012$	0.881	0.126	14	32	Calcite	Vaterite
CaCl <sub>2</sub> Concentration								
50 mM	$0.554 \pm 0.049$	$-0.464 \pm 0.025$	1.019	0.090	20	150	Calcite	–
10 mM	$0.694 \pm 0.023$	$-0.120 \pm 0.008$	0.814	0.225	10	13	Vaterite	Calcite
Solution Surface Area								
58 cm <sup>2</sup>	$0.685 \pm 0.034$	$-0.320 \pm 0.001$	1.034	0.176	10	40	Calcite	–
13 cm <sup>2</sup>	$0.139 \pm 0.007$	$-0.072 \pm 0.011$	0.223	0.050	33	10	Vaterite	Calcite
Sec. Diffusion Barrier Free Cross Sectional Area <sup>c)</sup>								
4 cm <sup>2</sup>	$0.089 \pm 0.003$	$-0.056 \pm 0.006$	0.156	0.063	35	40	Calcite	Vaterite
2 cm <sup>2</sup>	$0.070 \pm 0.014$	$-0.037 \pm 0.005$	0.107	0.014	47	50	Calcite	Vaterite
0.6 cm <sup>2</sup>	$0.025 \pm 0.006$	$-0.008 \pm 0.002$	0.0261	0.008	93	30	Vaterite	Calcite
Agitation								
100 rpm	$1.047 \pm 0.080$	$-0.324 \pm 0.018$	1.261	0.208	15	160	Calcite	Aragonite

<sup>a)</sup>Base condition (25 mM CaCl<sub>2</sub>, 70 mL, 48 cm<sup>2</sup>, 3 g (NH<sub>4</sub>)<sub>2</sub>CO<sub>3</sub>, 2.6 L overhead space, no secondary diffusion boundaries); <sup>b)</sup>Primary and secondary polymorph constituent after 100 min; <sup>c)</sup>200 min estimated based on SEM Images, Raman- and IR-Spectra.



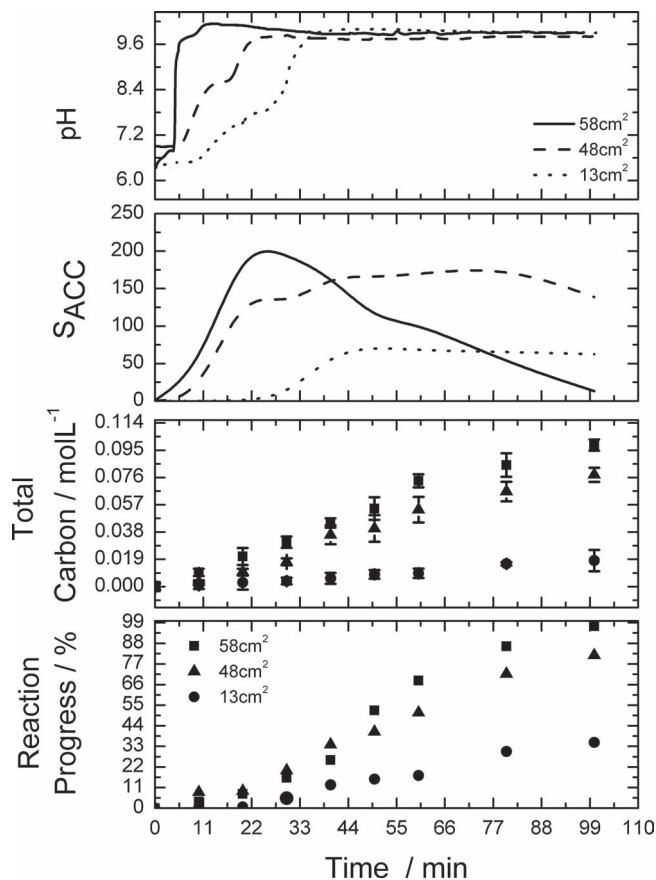
**Figure 4.** Graphs showing the influence of the initial calcium concentration on the induction point (as measured by the transmittance), the pH profile and the supersaturation. — 10 mM  $\text{CaCl}_2$ , - - 25 mM  $\text{CaCl}_2$ , - · - 50 mM  $\text{CaCl}_2$ . The data shown are an average of three experiments.

carbon addition rates,  $d[\text{C}]_{\text{T}}/dt$ , which were identical when either 1.5, 3, or 5 g of ammonium carbonate were employed (Table 1 and Supporting Information Figure S3). This result confirms that a near-constant vapor pressure is present.

### 2.3.2. Initial Calcium Concentration

The initial calcium concentration, by contrast, had some effect on the precipitation reaction, although the crystallization progress was little affected. Increasing the initial calcium concentration from 10 mM to 50 mM unexpectedly resulted in shifts in the induction point to longer times (**Figure 4**). This can be attributed to the corresponding retardation in pH increase (as shown in **Figure 4**) as the initial calcium concentration is increased, which in turn delays the conversion of bicarbonate to carbonate ions. This effect may be founded in the increased association of calcium ions with hydroxide ions originating from the ammonia to ammonium ion conversion.

The most significant effect of changing the calcium concentration was on the supersaturation. The supersaturation levels at induction decreased with decreasing calcium concentration due both to the lower calcium concentration and the lower levels of bicarbonate present at induction. On progression of the reaction, significantly higher supersaturation levels of  $\approx 230$  are reached in the 50 mM solution, as compared with  $\approx 75$  in the 10 mM solution, which can be associated with a higher nucleation density as the calcium concentration is raised.



**Figure 5.** Graphs of the influence of the solution surface area on the pH profile, supersaturation and total carbon added. ··· 13  $\text{cm}^2$ , - - 48  $\text{cm}^2$ , - - 58  $\text{cm}^2$ . The data shown are an average of three experiments.

Characterization of the reaction products sampled after 100 min using SEM and Raman microscopy showed that the Ca concentration also influenced the polymorph produced, and that a greater proportion of vaterite to calcite was obtained at 10 mM as compared with 50 mM (Supporting Information Figure S4 and **Figure S5**). The carbon addition rate also increased with the initial calcium concentration due to increased  $\text{CaCO}_3$  precipitation, resulting in an extended period of high  $\text{CO}_2$  diffusion into the solution, as shown in Table 1 and Supporting Information **Figure S6**.

### 2.3.3. Solution Surface Area

Variation of the solution surface area, while maintaining a constant volume, resulted in significant and systematic changes in all of the parameters investigated, indicating that this provides an excellent and straightforward method for controlling the diffusion process (**Figures 5** and Supporting Information **Figure S7**). Induction times decreased with increasing surface areas, while more rapid increases in the pH and associated nucleation rates were observed at higher surface areas. This is expected due to the more uniform supersaturation profiles in solutions with larger surface areas. **Figure 5** confirms the

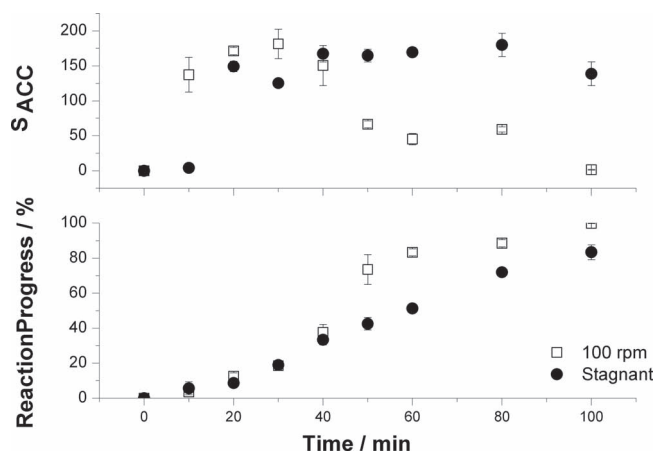
linearity of this relationship, which is in agreement with Fick's first law,<sup>[29]</sup> i.e., doubling the surface area leads to a twofold increase in carbon addition rate. Similarly, the reaction progress increased with increasing surface area, and the supersaturation peaked at earlier times in the more rapid (larger surface area) reactions. Therefore, larger surface areas are associated with shorter induction and overall reaction times, higher peak supersaturations, and nucleation rates. In contrast, the lower surface area provides conditions where supersaturations are maintained at constant values for long periods.

### 2.3.4. Diffusion Barriers

Introduction of an additional diffusion barrier (typically in the form of Parafilm perforated with needle-holes) is widely used as a method of regulating the ADM. Experiments were therefore conducted where the dish containing the reaction solution was covered with Parafilm punctured with 3, 10, or 20 holes, corresponding to a total free area of 0.6, 2, and 4 cm<sup>2</sup> respectively. A decrease in the rate of calcium loss and in the reaction progress was observed with a reduction in the free area, and the induction time increased from ≈15 min in the absence of a diffusion barrier to ≈100 min when there was only 0.6 cm<sup>2</sup> free area. However, in both of these cases, nucleation occurred at supersaturations above the ACC solubility limit (Table 1). The total carbon addition rate was found to exponentially decrease with a decrease in the free area, in contrast to the linear relationship recorded with respect to the liquid surface area. Thus, in common with a reduction in the solution surface area, reduction in the free area through introduction of a diffusion barrier leads to longer reaction times, where supersaturation remains at elevated levels for longer periods.

### 2.3.5. Stirring

The effect of agitation on the ADM was studied by addition of a magnetic stirrer to the CaCl<sub>2</sub> solution. An influence on the crystallization was only observed at stirring rates of 100 rpm and above, where a higher crystallization rate was recorded



**Figure 6.** Graphs showing the influence of solution agitation on the supersaturation and reaction progress. • Stagnant, □ 100 rpm. The data presented are averages over three experiments for a 25 mM CaCl<sub>2</sub> solution.

as compared with unstirred solutions (Figure 6). The effect of stirring was more pronounced in the later stages of crystallization, and no significant change in the induction time was observed, despite there being a more rapid initial increase in the pH (Supporting Information Figure S8). This is consistent with a faster build-up in ammonia and carbonate in the solution, and nucleation occurring at higher supersaturations in the stirred solutions. The increase in reaction rate in the later stages of the reaction can be explained by an increase in secondary nucleation events which occurs due to an increased frequency of particle collisions, the generation of additional nucleation sites through attrition processes, and the increased kinetic energy in the system. It is also supported by the presence of aragonite as well as calcite under stirred conditions, while only calcite is present under stagnant conditions (Supporting Information Figure S4). The formation of aragonite due to agitation has been reported elsewhere,<sup>[30]</sup> and may relate to the increase of kinetic energy in the system.

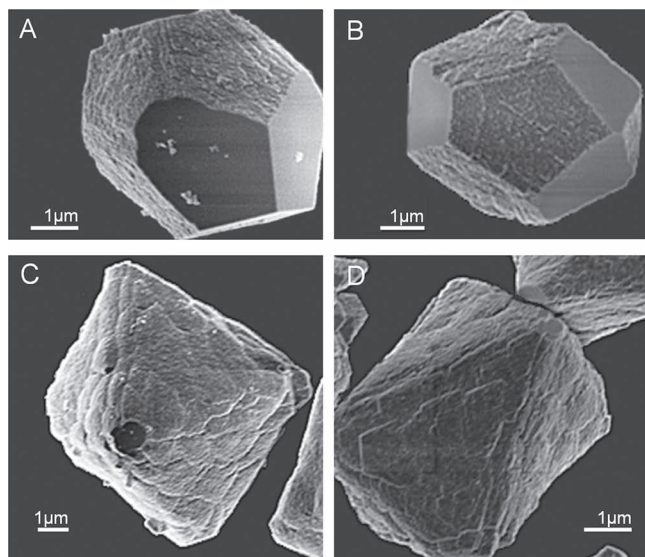
### 2.4. Reproduction of the Ammonia Diffusion Method

CaCO<sub>3</sub> mesocrystals formed in the presence of PSS-MA were chosen as a model system to test the translation of the obtained ADM reaction conditions to a titration-based system, due to their distinctive morphologies and properties (in particular their high surface areas of 60–100 m<sup>2</sup>/g). Success in the replication of mesocrystals is particularly important as it provides support for our analysis which suggests that a prolonged, steady supersaturation promotes the continuous formation of new material, which supports aggregation-based crystallization.

Here, a carbonate containing reagent solution (20 mL of either 250 or 175 mM (NH<sub>4</sub>)<sub>2</sub>CO<sub>3</sub>) was added at 0.0057 mL/min to a 80 mL of solution containing 325 ppm PSS-MA and either 5 mM or 1.25 mM CaCl<sub>2</sub>, under agitation at 70 rpm. The experimental conditions were selected to mimic the carbon addition rate during the crystallization zone (the first 100–200 min of reaction) of an ammonium carbonate diffusion reaction with the following conditions: 5 mM/1.25 mM CaCl<sub>2</sub>, 325 ppm PSS-MA, 70 mL, 48 cm<sup>2</sup>, and 0.6 cm<sup>2</sup> diffusion boundary pore surface area. The ammonium carbonate solution addition rate ( $k_S$ ) and concentration ( $[C]_{(NH_4)_2CO_3}$ ) added to a certain volume of calcium solution ( $V_{Ca}$ ) were determined according to Equation 4 such that they mimicked the carbon addition rate ( $d[C]_T/dt$ ) obtained experimentally from characterization of the ADM.

$$\frac{k_S [C]_{(NH_4)_2CO_3}}{V_{Ca} + k_L t} = \frac{d[C]_T}{dt} \quad (4)$$

Characterization of the product crystals demonstrated that they were calcite, as shown by Raman and IR spectroscopy (Supporting Information Figure S9), and that they were either pseudo-octahedral or dodecahedral in morphology (Figures 7A,B). Importantly, these morphologies were identical to those of the CaCO<sub>3</sub> mesocrystals produced using the ADM (Figures 7C,D). Further confirmation of mesocrystal structure was obtained by measurement of the surface areas of the crystals. Analysis of pseudo-octahedral mesocrystals produced by slow addition, and removed from the reaction solution after



**Figure 7.** SEM images of calcite crystals produced using the ADM and the slow addition technique. A) 1.25 mM Ca ADM, B) 1.25 mM Ca slow addition, C) 5 mM Ca ADM, and D) 5 mM Ca slow addition.

12 h, revealed typical surface areas of  $\approx 97$  m<sup>2</sup>/g. This value is consistent with the corresponding crystals produced using the ADM,<sup>[21]</sup> ( $\approx 82$  m<sup>2</sup>/g) and is considerably larger than the 1–2 m<sup>2</sup>/g recorded for rhombohedral calcite crystals of comparable sizes. Thermogravimetric analysis (TGA) of the “replica” mesocrystals revealed that they comprised 2–4 wt% polymer (Supporting Information Figure S10).

### 3. Discussion

A number of studies have previously attempted to either modify, or characterize features of the ammonia diffusion method in order to achieve greater reproducibility. These have included substitution of solid ammonium carbonate with a liquid reservoir of ammonium carbonate,<sup>[31,32]</sup> or estimation of pH and supersaturation changes (while neglecting precipitation) for precipitation in  $\mu$ L droplets in a so-called crystallization mushroom.<sup>[32]</sup> A general comparison of the ammonia diffusion and double diffusion methods has also been made by studying the pH profile.<sup>[33]</sup> By comparison, our approach provides a very detailed picture of the changes in solution which accompany the precipitation of CaCO<sub>3</sub> using the ADM.

The results presented here demonstrate that the ADM can be controlled using a range of variables including the introduction of a diffusion barrier, and change of the solution surface area, which leads to modification of the reaction profile. Where conditions are used which lead to a rapid reaction rate, the reaction profile and solution conditions approach those achieved in other techniques. This is characterized by a burst of nucleation which depletes a large proportion of the available calcium ions, followed by a steady drop in the supersaturation as the nuclei grow in solution. In contrast, when the ADM conditions are controlled to give slow growth, a unique profile can be generated

where nucleation occurs in an initial burst, which consumes only a relatively small proportion of the available ions, and the supersaturation then remains relatively constant, at a level well above the threshold for amorphous calcium carbonate (ACC) precipitation, until the calcium ions have been depleted. It is the precipitation of CaCO<sub>3</sub> crystals under the latter conditions that can lead to the generation of unique and often complex crystal morphologies when organic additives such as block copolymers are also present in the reaction solution.

Identification of the solution conditions which lead to these complex crystal structures—which determine the nucleation and growth processes which could feasibly occur—therefore provides a unique insight into the processes which may generate such crystals. As the precipitation occurs under conditions where the supersaturation remains rather constant, but at a level above the ACC threshold for the vast majority of the reaction, it is possible for new nuclei to form throughout the reaction, probably as ACC. This may occur homogeneously or heterogeneously on pre-existing precipitates in solution. If further nucleation does not occur, the nuclei formed in the original nucleation event will simply continue to grow. These nucleation and growth processes may of course also occur in tandem.

In the absence of polymer additives, calcite rhombohedra are the typical products of the ADM both under slow and rapid growth regimes. Studies of ACC precipitation in bulk solution (achieved by mixing solutions or calcium and carbonate ions) have shown that ACC particles form in a nucleation burst, and then continue to grow without aggregation.<sup>[34–36]</sup> The mechanism by which they crystallize is less clear. It has been suggested that nucleation of the later crystalline phases occurs within existing ACC particles,<sup>[37,38]</sup> and there is strong evidence that these crystalline nuclei then grow via dissolution/reprecipitation of other ACC particles in solution.<sup>[39–41]</sup> Further, an ACC particle cannot start to crystallize until it reaches a critical size.<sup>[42,43]</sup> Crystallization of ACC to vaterite in bulk solution via a solid state transformation has also been suggested, based on cryo-TEM techniques and SAXS/WAXS studies.<sup>[25,44]</sup> In the later study the ACC nanoparticles were believed to first dehydrate, then undergo a structural rearrangement to vaterite, and finally aggregate to form micrometer-scale vaterite particles.

This process will obviously be modified in the presence of polymers, which are likely to bind to the ACC particles, inhibiting their growth and potentially promoting aggregation. Some studies have attempted to characterize the mechanism of formation of CaCO<sub>3</sub> mesocrystals in the presence of polymer additives. These have analyzed the reaction solutions at early times using transmission electron microscopy (TEM), analytical ultracentrifugation and dynamic light scattering,<sup>[23,45,46]</sup> and have shown the presence of amorphous nanoparticles. As a particular feature of polymer-controlled growth, which distinguishes it from additive-free reactions, these ACC particles rapidly form aggregates, which then ultimately crystallize. Given the challenging nature of these early-time studies, they present only a broad picture of the reaction processes, and little is known about the growth of the aggregates or their crystallization mechanism.

Our results are therefore fully in keeping with these observations, but importantly also demonstrate that under slow growth in the ADM—which is the regime where mesocrystals

are observed—new material is continually produced after the initial nucleation of ACC. This then distinguishes it from rapid growth conditions where there is a single nucleation event, followed by growth. The most probable scenario is therefore that new particles nucleate on the existing polymer-stabilized ACC aggregates, or on crystalline particles at later stages of the reaction, giving rise to more complex morphologies.

The formation of  $\text{CaCO}_3$  mesocrystals would therefore appear to be based on the crystallization of an assembly of ACC nanoparticles rather than the oriented assembly of precursor crystalline nanoparticles as was originally suggested.<sup>[23,45]</sup> Indeed, nanoparticulate calcite and vaterite are very hard to synthesize due to their rapid growth in solution. In this way, synthetic  $\text{CaCO}_3$  mesocrystals would appear to have many similarities to biogenic calcite mesocrystals, where the ultrastructure derives from a memory of the ACC precursor phase.<sup>[13]</sup> It is also stressed that Ostwald ripening processes are active during  $\text{CaCO}_3$  precipitation, such that large crystals grow at the expense of smaller ones. Indeed, it is noted that under even the slowest reaction times used here, the reaction is complete and the calcium ions depleted after 6–8 h. Therefore, while a number of articles describe the precipitation of  $\text{CaCO}_3$  crystals using the ADM using prolonged incubation periods (days to weeks), any morphological changes reported in crystals after  $\approx 12$  h are simply due to Ostwald ripening/recrystallization processes. Given that polymer remains in the solution, rough crystal surfaces, as seen in  $\text{CaCO}_3$  mesocrystals would be expected.

## 4. Conclusions

Although the ammonium diffusion method is widely used to precipitate calcium carbonate, due in part to its ability to generate unique crystal morphologies, it is typically used as a “black box” with little or no understanding of the supersaturation levels or changes in solution which accompany crystallization. In this work, we have addressed this challenge and have provided a rigorous characterization of the ADM. This approach has enabled identification of the key solution conditions which give rise to these morphologies, and has shown that the distinguishing feature of the ADM is that the initial nucleation burst consumes only a relatively small amount of the available ions, and the supersaturation then remains relatively constant, and well above the threshold value for ACC, until the majority of the calcium ions have been consumed. New material is therefore generated throughout the course of the precipitation reaction, a feature which we believe to be fundamental to the formation of complex, aggregation-based morphologies. This understanding of the ADM enables control over the process, and has also facilitated development of a simple, slow addition-based synthesis method which can replicate  $\text{CaCO}_3$  mesocrystal morphologies. This new technique overcomes the inherent irreproducibility of the ADM, and provides great flexibility, enabling for example scale-up of the process, and the possibility of establishing a continuous crystallizer setup. Finally, although the focus of the study has been  $\text{CaCO}_3$ , it is expected that the solution conditions which promote aggregation-based crystal growth are likely to be quite general, such that it will be possible to use the insight gained here to control the crystallization of many materials.

## 5. Experimental Section

A full description of all methods and calculations used is provided in the Supporting Information, supported by an overview of key methods here.

*Characterization of the Ammonium Diffusion Method (ADM):* A crystallization dish containing calcium chloride solution was placed in a desiccator, and a pH electrode, a temperature recorder and tube through which sample aliquots could be removed were inserted into the solution (Figure 1). This dish was either uncovered, or was covered with perforated Parafilm to provide a secondary diffusion barrier. Solid ammonium carbonate, spread in a glass dish, was added to the chamber, which was then sealed. The changes in the reaction solution accompanying precipitation of  $\text{CaCO}_3$  were then evaluated by taking samples from the solution at key time points, and quenching them to prevent further reaction. The solution was then separated from the solid matter by centrifugation, and the total concentrations of aqueous calcium containing species and carbon present were determined using atomic absorption spectroscopy (AA) and ion chromatography (IC) respectively. The IC was performed using KOH as an eluent at pH 12, such that inorganic carbon species are purely in the form of carbonate ( $\text{CO}_3^{2-}$ ). The weight loss of the ammonium carbonate during the reaction was also determined. In a second set of experiments, the above setup was extended to determine the induction point by recording changes in the solution turbidity with time using UV-vis spectroscopy.<sup>[47]</sup>

*Calculation of Carbonate Concentration, Supersaturation and Crystallization Progress:* The solution carbonate concentrations ( $[\text{CO}_3^{2-}]$ ,  $[\text{HCO}_3^-]$  and  $[\text{H}_2\text{CO}_3]$ ) were calculated, based on the total inorganic carbon concentration ( $[\text{C}]_{\text{T5}}$ ) and pH measured,<sup>[26]</sup> using a simplified carbon mass balance and carbonic dissociation constants ( $K_A$ ). The crystallization progress ( $\xi$ ) was determined by recording the total calcium concentration in solution, as described below.<sup>[48]</sup>

$$\xi = \left( \frac{[\text{Ca}]_t - [\text{Ca}]_i}{[\text{Ca}]_t - [\text{Ca}]_{\text{equilibrium}}} \right)$$

The calculated supersaturation ( $S$ ) with respect to a specific polymorph ( $x$ ), was expressed as the ratio of ionic activity product to solubility product ( $K_{sp}$ ).

*Precipitation of Calcium Carbonate Mesocrystals:* In order to test whether the characterization of the ADM carbonate and supersaturation profiles could be translated to a diffusion-free experimental setup,  $\text{CaCO}_3$  mesocrystals were precipitated in the presence of poly(4-styrenesulfonic acid co-maleic acid) (PSS-MA) by mechanical slow addition of reagents into solution under conditions which paralleled the ADM. The crystal morphologies were then compared with those prepared from identical reagents using the ADM. Replication of the ADM was achieved by slow addition of a  $(\text{NH}_4)_2\text{CO}_3$  solution into a crystallization dish containing a  $\text{CaCl}_2$ -PSS MA solution using a syringe pump, where the system was under constant agitation with an orbital shaker. The addition rate and concentration of the  $(\text{NH}_4)_2\text{CO}_3$  were determined based on the carbon addition rates derived from the ADM. The surface area of the  $\text{CaCO}_3$  mesocrystals produced was determined using the Brunauer–Emmett–Teller (BET) method while thermogravimetric analysis (TGA) was applied to determine the polymer content in the crystals. Crystal morphologies were characterized using SEM and crystal polymorphs were determined by Raman microscopy. As a control,  $\text{CaCO}_3$  mesocrystals were also precipitated using the traditional ADM, carried out as described earlier.

## Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.



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